**Article ID**: 1007– 4627(2000) 03-0189-02

## Photoluminescences of SiO<sub>2</sub> Irradiated by Heavy Ion\*

WANG Zhi-guang<sup>1</sup>, JIN Yun-fan<sup>1</sup>, XIE Er-qing<sup>1,2</sup>, CHEN Xiao-xi<sup>1</sup>, ZHU Zhi-yong<sup>1</sup>, HOU Ming-dong<sup>1</sup>

(1 Institute of Modern Physics, the Chinese Academy of Sciences, Lanzhou 730000, China; 2 Physics Department, Lanzhou University, Lanzhou 730000, China)

**Abstract**: A novel method, high energy heavy ion irradiation, was proposed to produce light-emitting structures. It was used in studying photoluminescence properties from C-doped SiO<sub>2</sub> films.

Key words: high energy heavy ion irradiation; photoluminescence; SiO2 films

**CLC number:** O 571. 33 **Documber code:** A

Since the discovery of intense visible photoluminescence (PL) from porous Si at room temperature, there has been a great interest in the structural and optical properties Si nanocrystals<sup>[1]</sup>. A number of methods including chemical etching, CVD, co-sputtering, implantation have been used to produce lightemitting structures. However, most of them are not suitable for modern microelectronics application.

In this letter, a novel method, high energy heavy ion irradiation, was proposed to fabricate light-emitting materials with sufficient and special luminescent centers. The base of this method is that a very energetic heavy ion can induce defect production, ion latent track formation as well as phase change in a nanosized region along the ion path in a solid (see a review paper[2]). According to the effects of quantum confinement, chemical compounds and crystalline defects, quantum dots (point defects, nanosized ion track or new phase) created by high energy heavy ion irradiation will be light-emitting centers. Indeed, it is evidenced by the following experiments.

SiO<sub>2</sub> films were thermally-grown on (111) facet of p-type single crystalline silicon by use of wet oxidation at high temperature. The oxide thickness was about 500 nm. They were implanted

at room temperature (RT) with 120 keV C-ions to 5× 10<sup>17</sup> ions/cm<sup>2</sup>, and then irradiated at RT with 335 or 855 MeV <sup>40</sup>Ar ions or 1.98 GeV <sup>86</sup>Kr ions to 1 × 10<sup>12</sup> ions/cm<sup>2</sup>. After implantation and irradiation, spectra of PL excited by 320 nm monochromatic ultra-violet were measured at RT.

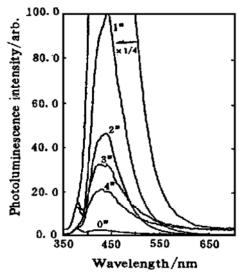


Fig. 1 Spectra of photoluminescence from SiO<sub>2</sub> samples. 0<sup>#</sup> as implanted; 1<sup>#</sup> as implanted and 855 MeV Ar ion-irradiated: 2<sup>#</sup> as implanted and 335 MeV Ar ion-irradiated; 3<sup>#</sup> as implanted and 1.98 GeV Kr ion-irradiated; 4<sup>#</sup> 1.98 GeV Kr ion-irradiated.

Fig. 1 shows the measured PL spectra. It is clear that blue-violet PL bands formed in the heavy ion irradiated samples ( $1^{\#} - 4^{\#}$ ). The most intense

Received date: 2000 -05 -23

<sup>\*</sup> Foundation item: NSFC (19875069); Scientific Foundation of "9. 5" Fundamental Research (KJ952-S1-423) and "Xibuzhiguang" from the Chinese Academy of Sciences

PL bands are centred at the region from 441 to 430 nm (2. 81 to 2. 88 eV). There is also a relative weak but detectable violet PL from 3<sup>#</sup> (peak centred at 384 nm, 3. 23 eV) and 4<sup>#</sup> (380 nm, 3. 26 eV) samples. For unirradiated C-doped SiO<sub>2</sub> thin film (0<sup>#</sup>), no blue PL peak but a weak violet PL peak centred at 380 nm (3. 26 eV) was observed. For 1. 98 GeV Kr-ion irradiations, the PL from the C-doped SiO<sub>2</sub> film (3<sup>#</sup>) is stronger than that from the thermally-grown SiO<sub>2</sub> film (4<sup>#</sup>).

These may be interpreted by ion latent track formation and new phase separation in SiO2 films under high energy heavy ion irradiations. Meftah et al<sup>[3]</sup> have found that there is a threshold electronic energy loss (S<sub>e</sub>) value (S<sub>et</sub>≈ 1.6 keV/ nm) for ion track formation in SiO<sub>2</sub>. If  $S_e < S_{et}$ , the incident ion could only produce point defects in  $SiO_2$ . When  $S_e > S_{et}$ , ion track could be produced and its size increases with increasing the  $S_e$  value. For 855 MeV Ar-ion irradiation (1<sup>#</sup>,  $S_e = 1.48$  $keV/nm < S_{et}$ ), the PL band centred at ~ 440 nm is due to defect formation or bond sessions<sup>[4]</sup>. For 335 MeV Ar or 1.98 GeV Kr ion irradiations (S<sub>e</sub>>  $S_{\rm et}$ ), nanosized ion track (new phase) can form. At this condition, the measured PL band and its peak position are determined by the competition of PL from the ion track and the point defects outside

the track. The appearance of ion track results in shift of blue PL peak position towards to the shortwavelength direction and weakeness of the PL intensity from the samples. Taking into account the energy of the Si-Si, Si-C and C-C bands (2.36, 3.21 and 3.70 eV) and the maximum atomic concentration of implanted C-ions (~ 4× 10<sup>22</sup> C-atoms/cm<sup>3</sup>) at the projected range, the relative weak PL bands (peak centred at about 380 nm) may be due to phase separation into SiC and carbon aggregates along the ion path. Moreover, the existence of doped C-atoms will stabilize the produced point defects and enhance luminescence yield.

Experimental results let us make the conclusion that the high energy heavy ion irradiation is a useful and effective technique for fabricating luminescence materials and, in our knowledge, it is the first time to be proposed and used to obtain PL materials. PL from ion irradiated C-doped SiO<sub>2</sub> films may mainly due to the quantum-confined effect and crystalline defects.

We wish to thank the staff members of the HIRFL and the 200 kV implantor of Institute of Modern Physics for their helps during implantation and irradiation experiments.

## **References:**

- [1] Cullis A G, Canham L T, Calott D J. The Structural and Luminescence Properties of Porous Silicon [J]. J Appl Phys, 1997, 82(3): 909-965.
- [2] Wang Z G, Jin Y F, Hou M D. Experimental Study of Electronic Energy Loss Effects in Pure Metals [J]. Nucl Phys Rev (in Chinese), 2000, 17(2): 100-105.
- [3] Meftah A, Brisard F, Costantini J M et al. Track Formation in SiO<sub>2</sub> Quartz and the Thermal-spike Mechanism [J]. Phys Rev, 1994, B49: 12 457- 12 463.
- [4] Shluger A, Stefanovich E. Models of the Self-traped Exciton and Nearest-neighbor Defect Pair in SiO<sub>2</sub>[J]. Phys Rev, 1990, B42: 9 664- 9 673.

## 重离子辐照 SiO<sub>2</sub> 的光致发光\*

王志光<sup>1</sup>, 金运范<sup>1</sup>, 谢二庆<sup>1,2</sup>, 陈晓曦<sup>1</sup>, 朱智勇<sup>1</sup>, 侯明东<sup>1</sup> (1中国科学院近代物理研究所, 甘肃 兰州 730000; 2 兰州大学物理系, 甘肃 兰州 730000)

摘 要: 基于荷能离子与固体相互作用特点,提出了一种新的制备光致发光材料的方法—— 高能重离子辐照. 用这种方法研究了 SiO<sub>2</sub> 薄膜的光致发光特性,发现高能<sup>84</sup>Kr 和<sup>40</sup>Ar 离子辐照可在注碳 SiO<sub>2</sub> 薄膜样品中产生强的蓝-紫光发射带,掺杂碳增强了辐照样品的发光特性.

<sup>·</sup> 基金项目: 国家自然科学基金资助项目(19875069); 中国科学院"西部之光"基金资助; 中国科学院"九五"基础性研究重点基金资助项目(KJ952-S1-423)